

## CHAPTER V

### MISCIBILITY AND SPECIFIC INTERACTION BETWEEN 17 $\beta$ -ESTRADIOL AND EUDRAGIT<sup>®</sup> RS SOLID DISPERSION DETERMINED BY MTDSC AND FTIR

#### 5.1 Introduction

A monolithic matrix is the simplest system used to control drug release. The system is composed of drug homogeneously distributed in the polymer or other carrier materials (Kim, 2000a). Solid dispersion techniques, i.e. fusion method or solvent evaporation method, are generally used to distribute drug into the polymeric matrix (Serajuddin, 1999). Miscibility of drug in the polymer is an important factor indicating the uniformity of drug in the polymeric matrix either molecular dispersion in a dissolved state or regular crystallization in a crystalline state. If drug and polymer are immiscible, lack of uniformity of drug dispersion and irregular drug crystallization in the polymeric matrix will be occurred (Greenhalgh et al., 1999). Moreover, drug release from the polymeric matrix will be varied.

In polymer sciences, the miscibility of a polymer blend composed of two amorphous polymers can be determined by thermal analysis. A single  $T_g$  point is an indication of full miscibility of a polymer blend (Kuo and Chang, 2001; Kuo et al., 2001; Maldonado-Santoyo et al., 2004). Otherwise an immiscible polymer blend exhibits more than one  $T_g$  representing  $T_g$  of amorphous pure polymers used as components (Kuo and Chang, 2001). For most polymer blends,  $T_g$  of a miscible blend lies between  $T_g$  of the pure polymers and can be predicted by Gordon-Taylor equation or its modified version equation (Maldonado-Santoyo et al., 2004). Weight fractions and  $T_g$  of pure polymer components are major factors in predicting  $T_g$  of miscible polymer blends by Gordon-Taylor equation according to earlier described equation (2.16)

$$T_g = \frac{w_1 T_{g1} + K w_2 T_{g2}}{w_1 + K w_2} \quad 2.16$$

and  $K$  can be expressed in accordance with the Simha-Boyer rule;

$$K = \frac{\rho_1 T_{g1}}{\rho_2 T_{g2}} \quad 2.17$$

where  $T_{gi}$ ,  $w_i$ , and  $\rho_i$  are the glass transition temperature, the weight fraction and the density of the pure polymer components, respectively (Schneider, 1988). In case of a polymer blend exhibiting an interaction between the pure polymer components, Gordon-Taylor equation fails to predict the  $T_g$ . Kwei equation, a modified version of Gordon-Taylor equation, has been used to predict the  $T_g$  of this

type of polymer blend (Maldonado-Santoyo et al., 2004; Schneider, 1988). Kwei equation is written as formerly described equation (2.27);

$$T_g = \frac{w_1 T_{g1} + w_2 T_{g2}}{w_1 + K w_2} + q w_1 w_2 \quad 2.27$$

$q$ , in the additional term, is adjustable parameter corresponding to the strength of hydrogen bonding in a polymer blend (Kuo and Chang, 2001; Kuo et al., 2001; Maldonado-Santoyo et al., 2004).

In pharmaceutical system, several drugs in a dissolved state in polymeric matrices were reported as existing in amorphous form (Satit Puttipipatkachorn et al., 2001; Jenquin et al., 1990; Jenquin et al., 1992). This system is nearly identical with the amorphous polymer blend. Furthermore, many solid solutions of drugs in the polymeric matrices such as salicylic acid or chlorpheniramine in ERL or ERS films have exhibited the decrease of  $T_g$  when the concentrations of those in the polymeric films increase (Jenquin et al., 1990; Jenquin et al., 1992). These phenomena have been similar to those of ibuprofen or methylparaben in ERS 30D films (Wu and McGinity, 1999; Wu and McGinity, 2003). The Mixture of  $17\beta$ -estradiol ( $E_2$ ) in polyvinylpyrrolidone (PVP) after passing melt extrusion process has also exhibited the increase of  $T_g$  toward  $T_g$  of PVP when the concentration of PVP increases (Hülsmann et al., 2001). Few researches in pharmaceutical field have been done to explain the miscibility of drug in polymer based on the principle of Gordon-Taylor equation.

For polymer blends composed of a crystalline polymer and an amorphous polymer, the criteria for indicating the miscibility have been used not only the variation of  $T_g$  as a function of composition but also the melting point depression. The reduction of melting temperature as a function of composition and polymer-polymer interaction parameter can be analyzed by Nishi-Wang equation (Nishi and Wang, 1975) based on the Flory-Huggins theory. Nishi-Wang equation can be written as early mentioned equation;

$$T_m - T_{mb} = \frac{-T_m B V_2 \phi_1^2}{\Delta h_2} \quad 2.43$$

where  $T_m$  and  $T_{mb}$  are melting temperatures of the crystalline polymer and the blend, respectively;  $B$  is the interaction energy density between blend components;  $V_2$  is the molar volume of the repeating unit of the crystalline polymer;  $\phi_1$  is the volume fraction of the amorphous polymer in the blend;  $\Delta h_2$  is the heat of fusion of the crystalline polymer per mole of the repeating unit (Rostami, 2000; Pimbert et al., 2002).

The polymeric matrix containing the concentration of drug higher than its solubility in polymer exhibits the crystallization of drug in the polymeric matrix as if the blend of a crystalline polymer with an amorphous polymer. In addition, piroxicam in PVP solid dispersion has exhibited melting point depression when the proportion of

PVP increases (Tantishaiyakul et al., 1996). This phenomenon has also been found in salicylic acid in chitosan solid dispersion (Satit Puttipipatkachorn et al., 2001). Few researches in pharmaceutical sciences have explained the reduction of melting point of drug in the polymeric matrix based on the Flory-Huggins theory. Pharmaceutical research reports have rarely been analyzed melting point depression and drug-polymer interaction parameter by using Nishi-Wang equation based on the Flory-Huggins theory.

In case of the interaction between drug and polymer occurring, the inter-associated hydrogen bond between drug and polymer should occur and can also be investigated by fourier transform infrared (FTIR) spectroscopy (He, Zhu, and Inoue, 2004).

The objectives of this study were

(i) to characterize polymorphic forms of  $E_2$  by thermal analysis and XRPD. Although the different polymorphic forms of  $E_2$  have been observed by thermal analysis, X-ray diffraction pattern of each polymorphic form of  $E_2$  has not been elucidated. This work attempted to characterize X-ray diffraction pattern of each polymorphic form of  $E_2$ .

(ii) to investigate miscibility of  $E_2$  and ERS in solid dispersion by using the principle of Gordon-Taylor equation and the Flory-Huggins theory. If the miscibility can be forecasted by these proposed theories, thermal analysis will be one of various techniques for determining the miscibility of  $E_2$  and ERS.

(iii) and finally to examine the specific interaction between  $E_2$  and ERS by FTIR for supporting the prediction of the miscibility by these proposed principles.

The miscibility of  $E_2$  in ERS indicates not only how much the possibility to achieve the uniformity of  $E_2$  in ERS solid dispersion but also how much the potential for using ERS as a release controlling agent for  $E_2$ .

## 5.2 Materials and Methods

### 5.2.1 Materials

17 $\beta$ -estradiol (E<sub>2</sub>) was purchased from Fluka Chemica, Germany. Eudragit® RS PO (Röhm Pharma GmbH, Germany) was supplied as a gift by JJ Degussa, Thailand. Absolute ethanol was of a reagent grade purchased from Merck, Germany. The chemical structures of E<sub>2</sub> and ERS are shown in Figure 5.1.

### 5.2.2 Preparation of E<sub>2</sub> in ERS Solid Dispersions

Solid dispersions containing E<sub>2</sub> in ERS at concentration range of 1-90 % w/w were prepared by solvent evaporation. E<sub>2</sub> and ERS at specific weight ratios were dissolved in the minimum volume of absolute ethanol to get clear solution and then poured into Teflon coated plate (15.5 cm x 15.5 cm). The absolute ethanol was evaporated at room temperature. Dried samples were kept in a desiccator over silica beads for further use.

### 5.2.3 Physico-Chemical Characterization of E<sub>2</sub>, ERS, and E<sub>2</sub> in ERS Solid Dispersions

#### 5.2.3.1 Differential Scanning Calorimetry (DSC)

DSC was carried out using a TA Q100 with a refrigerated cooling system (TA Instruments, New Castle, DE) and nitrogen as purge gas. The calorimeter was calibrated using indium and sapphire for temperature and heat capacity, respectively. E<sub>2</sub> of 3.17 mg was added to standard aluminum pan with cover (closed pan) and scanned using the following heating program: heating up to 182 °C at 10 °C/min; cooling down to 0 °C at 20 °C/min; heating up to 250 °C at 10 °C/min. This heating program enabled water loss. T<sub>g</sub> and melting points of each form of E<sub>2</sub> to be determined in a run.

#### 5.2.3.2 Thermogravimetric Analysis (TGA)

TGA was carried out using a Mettler Toledo TGA apparatus (TGA/SDTA 851e, Switzerland). E<sub>2</sub> was added in open pan and scanned from 0-750 °C at heating rate of 5 °C/min. Weight loss was determined gravimetrically. This technique was used to investigate the change of E<sub>2</sub> mass as a function of temperature in order to support the events occurring in the DSC curve.

#### 5.2.3.3 X-ray Powder Diffraction Analysis (XRPD)

XRPD was carried out using a Siemens D5000 diffractometer (Stuttgart, Germany). E<sub>2</sub>, E<sub>2</sub> heated from 0-175 °C at 5 °C/min, E<sub>2</sub> heated from 0-180.5 °C at 5 °C/min, and E<sub>2</sub> heated from 0-180.5 °C at 5 °C/min, cooled down to 0 °C at 20 °C/min, finally heated up to 140 °C at 5 °C/min were scanned from 5° to

90° 2 $\theta$  (sampling interval of 0.02°) using Ni-filtered Cu K $\alpha$  radiation. Operating voltage and current were 40 kV and 30 mA, respectively.

#### 5.2.3.4 Modulated Temperature Differential Scanning Calorimetry (MTDSC)

MTDSC was carried out using a TA Q100 Modulated DSC with a refrigerated cooling system (TA Instruments, New Castle, DE) and nitrogen as purge gas. E<sub>2</sub>, ERS, and E<sub>2</sub> in ERS solid dispersions at concentration range of 1-90 % w/w of each were added to standard aluminum pans with covers (closed pan) and scanned using two heating programs. Heating program I: heating from 25 to 120 °C at 10 °C/min, cooling down to 25 °C at 20 °C/min, an isothermal period for 5 min at 25 °C, and finally heating to 250 °C at 5 °C/min. Heating program II: heating from 25 to 182 °C at 10 °C/min, cooling down to 25 °C at 20 °C/min, an isothermal period for 5 min at 25 °C, and finally heating to 250°C at 5 °C/min. A modulation amplitude of  $\pm 1^\circ\text{C}$  and a period of 60 s were used. MTDSC was employed to determine T<sub>g</sub> and E<sub>2</sub> melting point in solid dispersions because of the advantage of MTDSC over DSC for distinguishing overlapping thermal events of E<sub>2</sub> and ERS.

#### 5.2.3.5 Fourier Transform Infrared Spectroscopy (FTIR)

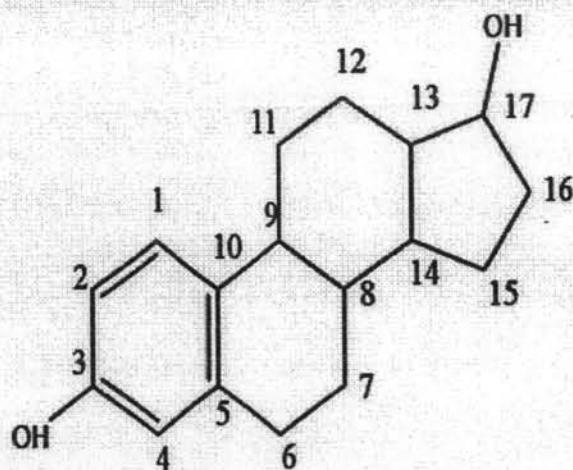
FTIR spectra of ERS, E<sub>2</sub> and solid dispersions containing 20, 50 and 75 % w/w E<sub>2</sub> in ERS both not heated and heated from 0-175 °C at 5 °C/min were performed with a Perkin-Elmer FTIR Spectrum One. Potassium bromide disks were prepared by grinding 2-3 mg of the respective samples with 150 mg of dried potassium bromide. Disks of 13 mm in diameter were compressed at a pressure of 6000 Psi for 3 min. The disks were then mounted in a FTIR case and scanned from 4000 to 450 cm<sup>-1</sup>. Spectrometer adjustments were: resolution of 4 cm<sup>-1</sup> and sample scan of 64 times. This technique was used to determine the hydrogen bonding between E<sub>2</sub> and ERS in solid dispersions.

### 5.2.4 Mathematical Analysis

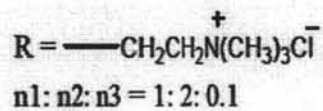
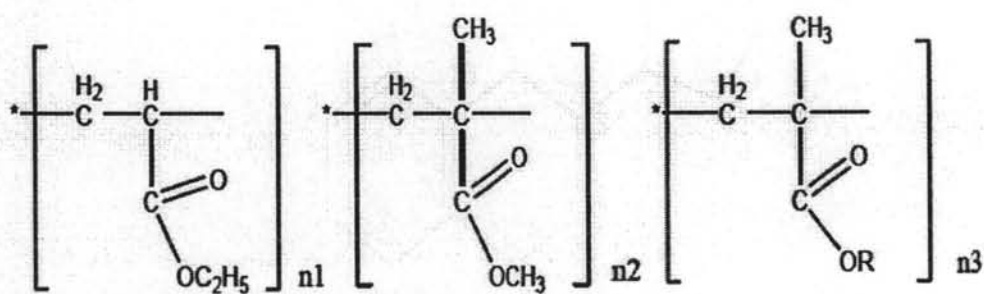
#### 5.2.4.1 Melting Point Depression Analysis

The Nishi-Wang equation was fitted to the melting point of E<sub>2</sub> in ERS solid dispersions determined by MTDSC in the concentration range of 20-100 % w/w. The B value was estimated by fitting the Nishi-Wang equation to experimental data by nonlinear regression analysis (GraphPad Prism<sup>®</sup> version 4.0).  $\Delta h_2$  (143.2 J/g) and T<sub>m</sub> (179.88 °C) were obtained from the MTDSC thermogram of pure E<sub>2</sub>. V<sub>2</sub> (168.99 cm<sup>3</sup>) was calculated by summation of the volumes of the structural groups of E<sub>2</sub> (VanKrevelen, 1990). Density of each component was obtained from the ratio of molecular weight to molar volume. Volume fraction of ERS ( $\phi_1$ ) was calculated from the weight fractions and densities of the components. Nine experimental data points were used for this fit. The coefficient of determination (R<sup>2</sup>) and randomness of the residuals were used to determine the goodness of fit.

(A)



(B)



**Figure 5.1** Chemical structure of  $\text{E}_2$  (A) (Barnett et al., 1995; Variankaval et al., 2000; Park et al., 2005) and ERS (B) (Lehmann, 1997)

### 5.2.4.2 T<sub>g</sub> Analysis

Using nonlinear regression, the Gordon-Taylor and Kwei equations were fitted to the T<sub>g</sub> versus weight fraction of ERS (w<sub>1</sub>) data obtained from the MTDSC thermograms of E<sub>2</sub> in ERS solid dispersions (0-100 % w/w). T<sub>g1</sub> and T<sub>g2</sub> were obtained from MTDSC thermograms of ERS and E<sub>2</sub>, respectively. K was estimated from the Gordon Taylor fit. K and q were estimated from the Kwei fit. Thirteen experimental data points were used for each fit. The coefficient of determination (R<sup>2</sup>) and the residual plot were used to evaluate the goodness of each fit. The best model was selected on the basis of the Akaike Information Criterion (AIC).

The AIC is a measurement of goodness of fit based on maximum likelihood. When comparing several models for a given set of data, the model associated with the smallest value of AIC is regarded as giving the best fit out of that set of models. The AIC value can be calculated from the below relationship;

$$AIC = n * \ln\left(\frac{SSR}{n}\right) + 2p \quad 5.1$$

where n is the number of data points, p is the number of the estimated parameters of the model, and SSR is the residual sum of squares (Buckland, Burnham, and Augustin, 1997; Costa and Lobo, 2001).

## 5.3 Results and Discussion

### 5.3.1 DSC and TGA Curve of E<sub>2</sub>

It has been reported that the most common form of E<sub>2</sub> is in hemihydrate state which is assigned to EA form according to the Variankaval et al. study (Variankaval et al., 2000). In addition, it can exist in the amorphous form and different polymorphic modifications. It is shown that crystal forms of E<sub>2</sub> are interconvertible by various thermal conditions. As illustrated in Figure 4.4, three endothermic peaks were observed in DSC curve of E<sub>2</sub> when scanned by the first heating run (DSC; 25-182 °C at 10 °C/min). The first and the second endothermic peaks were observed around 110 °C and 174 °C, respectively. These two endothermic peaks corresponded to the weight loss around 3.20 % according to the TGA curve of E<sub>2</sub> as illustrated in Figure 4.5. This observed water loss corresponded to the stoichiometry of E<sub>2</sub> given as C<sub>18</sub>H<sub>24</sub>O<sub>2</sub>·½ H<sub>2</sub>O which was in agreement with Variankaval et al. (2000) study. Variankaval et al. (2000) indicated that the water loss was composed of two stages. The first water loss peak after 110 °C corresponded to a partial release of hydrogen bonded water. The second peak at 174 °C corresponded to the complete loss of lattice water. The third endothermic peak at 179.24 °C was the melting point of E<sub>2</sub>. Therefore, the DSC curve obtained from the first heating run was identical to the DSC curve of estradiol hemihydrate.

The DSC curve during cooling down (DSC; 182-0 °C at 20 °C/min) exhibited  $T_g$  of  $E_2$  around 84.20 °C corresponding to  $T_g$  of  $E_2$  observed in the second heating run (DSC; 0-250 °C/min at 10 °C/min). This indicated that  $E_2$  was changed to an amorphous form after heated to 182 °C and rapidly cooled down because of manifesting  $T_g$ . In the second heating run, the amorphous form of  $E_2$  in the glassy state was changed to the rubbery state when the temperature was higher than 80 °C. Exothermic peak in DSC curve was observed around 126 °C followed by two endothermic peaks at 169.56 °C and 179.24 °C, respectively. The first endothermic peak was lower than the second one about 10 °C. The second endothermic peak was in the same position as the third endothermic peak of the first heating run. This thermogram agreed with the result obtained by Variankaval et al. (2000) indicating that the first endothermic peak and the second endothermic peak in the second heating run were attributed by other two forms of  $E_2$  which were assigned as ED form and EC form, respectively. It was suggested that the amorphous form of  $E_2$  under the rubbery state crystallized to ED form and EC form at 126 °C. EC form of  $E_2$  was in an anhydrous form because the lattice water was completely lost after the first heating run. The endothermic peak at 169.56 °C corresponding to ED form of  $E_2$  suggested that ED form manifested itself as a mixture with EC form.

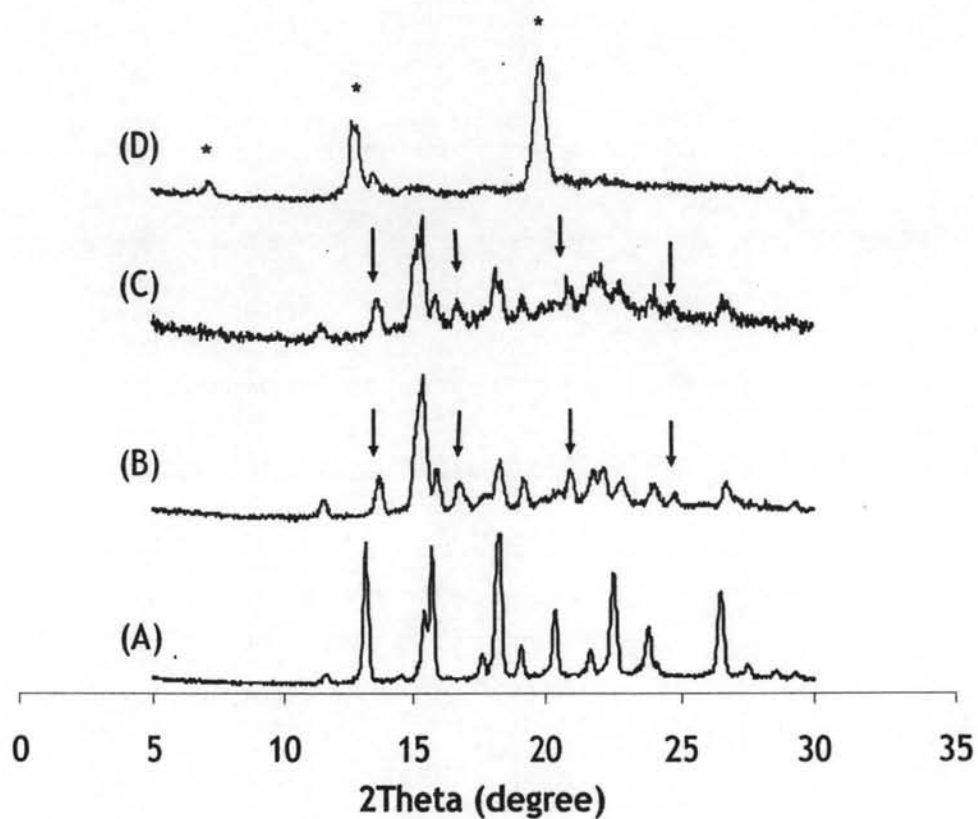
### 5.3.2 X-ray Powder Diffraction Pattern of $E_2$

X-ray powder diffraction patterns of  $E_2$  subjected to different heat treatments are shown in Figure 5.2.  $E_2$  without heat treatments corresponding to hemihydrate form exhibited characteristic peaks at a diffraction angle of  $2\theta$ , at 13.14, 15.74, 18.26, 22.62, and 26.58°. These diffraction angle and diffraction pattern were similar to those published by Latsch et al. (2003) and Park et al. (2005). The peaks at a diffraction angle of  $2\theta$ , at 13.60, 16.78, 21.00, and 24.72° were observed in a diffractogram of  $E_2$  heated from 0-175 °C and 0-180.5 °C at heating rate of 5 °C/min but these peaks were not observed in a diffractogram of  $E_2$  hemihydrate. The diffraction pattern of  $E_2$  heated from 0-180.5 °C was nearly identical with that of  $E_2$  heated from 0-175 °C. However, the relative peak intensity of a diffraction pattern of  $E_2$  heated from 0-180.5 °C was less than that of  $E_2$  heated from 0-175 °C and the peaks of a diffraction pattern of  $E_2$  heated from 0-180.5 °C did not significantly separate from baseline. It was a sign of a mixture of crystalline form and amorphous form. This suggested that  $E_2$  hemihydrate incompletely changed to the amorphous form after heated until the temperature reached to 180.5 °C but the lattice water was completely lost. Therefore, some portions of hemihydrate form of  $E_2$  changed to the amorphous form and some portions of hemihydrate form of  $E_2$  changed to the anhydrous form.

The diffraction pattern of  $E_2$  heated from 0-180.5 °C at 5 °C/min and cooled down to 0 °C at 20 °C/min and finally heated to 140 °C at 5 °C/min was different from the others and exhibited the characteristic peaks at a diffraction angle of  $2\theta$ , at 12.64 and 19.94°. The information obtained from DSC analysis indicated that crystallization was completely occurred when the temperature reached over 140 °C and ED form was firstly obtained in the second heating run. Therefore, a diffraction pattern of  $E_2$  obtained from this heat treatment should belong to ED form. The results



obtained from XRPD analysis confirmed the detection of three crystalline forms of E<sub>2</sub> as previously discussed.



**Figure 5.2** XRPD of E<sub>2</sub> treated with different heat treatments at heating rate of 5 °C/min: (A) no heat treatment; (B) 0-175 °C; (C) 0-180.5 °C; (D) 0-180.5 °C, 180.5-0 °C, 0-140 °C

### 5.3.3 MTDSC Analysis of E<sub>2</sub> in ERS Solid Dispersions with Heating Program I

T<sub>g</sub>, melting point, and heat of fusion of E<sub>2</sub> in ERS solid dispersions at concentration range of 0-90 % w/w determined by MTDSC at heating program I are presented in Table 5.1. Pure E<sub>2</sub> exhibited melting point at 179.88 °C. The melting point and heat of fusion of E<sub>2</sub> correspondingly decreased when the concentration of ERS in solid dispersion increased. This result revealed the melting point depression as a function of composition of E<sub>2</sub> in ERS solid dispersion. It indicated that the miscibility of E<sub>2</sub> and ERS was found in the molten state. The addition of ERS as the amorphous component into the blends decreased the chemical potential of the crystalline E<sub>2</sub> and led to a reduction of melting point.

To investigate T<sub>g</sub> of pure ERS and solid dispersions of E<sub>2</sub> in ERS, T<sub>g</sub> was analyzed from reverse heat flow in the second heating run, MTDSC; 25-250 °C, 5 °C/min. T<sub>g</sub> of pure ERS was found to be 68.38 °C. Each solid dispersion of E<sub>2</sub> in ERS at the concentration range of 1-90 % w/w exhibited only one T<sub>g</sub> which was not significantly different from the T<sub>g</sub> of pure ERS. In case of miscibility of E<sub>2</sub> and ERS, T<sub>g</sub> of E<sub>2</sub> in ERS solid dispersions should lie between T<sub>g</sub> of E<sub>2</sub> (84.20 °C) and T<sub>g</sub> of ERS (68.38 °C) as a function of composition. This suggested that the resulting T<sub>g</sub> of solid dispersions obtained from heating program I was the T<sub>g</sub> of ERS. T<sub>g</sub> values of E<sub>2</sub> in ERS solid dispersions were not altered during measurement by MTDSC using heating program I. From DSC curve of E<sub>2</sub> in Figure 4.4, E<sub>2</sub> changed to the amorphous form when heated over its melting point and rapidly cooled down. Only E<sub>2</sub> in the amorphous form exhibited T<sub>g</sub> because of its amorphous nature. In heating program I, E<sub>2</sub> was still in the crystalline form blended with ERS. Therefore, measuring T<sub>g</sub> of E<sub>2</sub> in ERS solid dispersions by MTDSC at heating program I was the measurement of T<sub>g</sub> of ERS.

### 5.3.4 MTDSC Analysis of E<sub>2</sub> in ERS Solid Dispersions with Heating Program II

T<sub>g</sub> values of E<sub>2</sub> in ERS solid dispersions at concentration range of 0-100 % w/w obtained from MTDSC using heating program II are shown in Figure 4.6 (b). T<sub>g</sub> of pure ERS, E<sub>2</sub> in ERS solid dispersions, and pure E<sub>2</sub> were determined in the second heating run, MTDSC; 25-250 °C, 5 °C/min. Pure ERS and pure E<sub>2</sub> manifested their T<sub>g</sub> at 66.21 °C and 83.77 °C, respectively. E<sub>2</sub> in ERS solid dispersions at concentration range of 1-90 % w/w manifested a single T<sub>g</sub> lying between those of ERS and E<sub>2</sub> as a function of composition. Furthermore, T<sub>g</sub> of E<sub>2</sub> in ERS solid dispersions shifted toward T<sub>g</sub> of E<sub>2</sub> as weight fractions of E<sub>2</sub> increased. This suggested that E<sub>2</sub> was transformed to an amorphous state after the first heating run causing alteration of T<sub>g</sub> of E<sub>2</sub> in ERS solid dispersions and the change of T<sub>g</sub> values depended on the amount of amorphous E<sub>2</sub> and ERS. This drug-polymer blend was identical with the amorphous polymer blend. Therefore, T<sub>g</sub> behavior of E<sub>2</sub> in ERS solid dispersion followed the principle of Gordon-Taylor equation in case of the miscibility.

**Table 5.1 Glass transition temperatures ( $T_g$ ), melting points ( $T_{mb}$ ), and heat of fusions ( $\Delta h$ ) of  $E_2$  in ERS solid dispersions at concentration range of 0-100 % w/w obtained from reverse heat flow of MTDSC scanning with the heating program I**

Substance	$T_g$ ( $^{\circ}C$ )	$T_{mb}$ ( $^{\circ}C$ )	$\Delta h$ (J/g)
ERS	68.38	-	-
1% $E_2$ in ERS	68.48	-	-
2% $E_2$ in ERS	68.49	-	-
10% $E_2$ in ERS	68.99	-	-
20% $E_2$ in ERS	68.53	132.10	0.0860
30% $E_2$ in ERS	68.50	147.31	1.0930
40% $E_2$ in ERS	69.60	153.35	5.1840
50% $E_2$ in ERS	69.44	159.65	15.7300
60% $E_2$ in ERS	68.07	168.73	28.3000
75% $E_2$ in ERS	70.27	172.25	42.0500
80% $E_2$ in ERS	69.23	175.09	53.0500
90% $E_2$ in ERS	68.94	178.46	129.9000
$E_2$	-	179.88	143.2000

### 5.3.5 Melting Point Depression Analysis

Nishi-Wang equation was used to predict the melting point depression of E<sub>2</sub> in ERS solid dispersion. The determination of B value for E<sub>2</sub> in ERS solid dispersion by fitting Nishi-Wang equation to experimental data is displayed in Figure 5.3. The measured melting point of E<sub>2</sub> in solid dispersion (T<sub>mb</sub>) was plotted against the volume fraction of ERS in solid dispersion (Ø<sub>1</sub>). The volume fraction was obtained from converting the weight fraction into volume fraction using the molar volume of monomeric units determined by the group contribution method (VanKrevelen, 1990). The fit of Nishi-Wang equation to experimental data exhibited good agreement between predicted T<sub>mb</sub> and experimental T<sub>mb</sub>. The B value obtained from curve-fitting was equal to -0.2808 J/(g\*cm<sup>3</sup>). This indicated the miscibility of E<sub>2</sub> and ERS in solid dispersion and specific interaction between E<sub>2</sub> and ERS in molten state. The coefficient of determination (R<sup>2</sup>) was 0.9804 which confirmed a good agreement between predicted T<sub>mb</sub> and experimental T<sub>mb</sub>. The distribution of the resulting residuals corresponding to the fit was plotted as shown in Figure 5.4. A random distribution was presented. This indicated a validity of Nishi-Wang equation to predict T<sub>mb</sub> of E<sub>2</sub> in ERS solid dispersion.

### 5.3.6 T<sub>g</sub> Analysis

Gordon-Taylor equation and its modified version equation, Kwei equation, were used to predict T<sub>g</sub> of E<sub>2</sub> in ERS solid dispersion. The curve-fitting of these two equations to experimental data is presented in Figure 5.5. The measured T<sub>g</sub> of E<sub>2</sub> in solid dispersion was plotted against weight percent of ERS in solid dispersion. The fit of Gordon-Taylor equation and Kwei equation to experimental data gave R<sup>2</sup> equal to 0.8932 and 0.9500, respectively. These values indicated that the predicted T<sub>g</sub> values obtained from Kwei equation were better agreement with the experimental data than those obtained from Gordon-Taylor equation.

The residuals corresponding to the fit of Gordon-Taylor equation and Kwei equation to experimental data are shown in Figure 5.6 and 5.7, respectively. The residuals corresponding to the fit in Figure 5.6 and 5.7 seemed to be a kind of structure in the distribution. This was an indication of a small deviation between theory and experiment. In order to proof the validity of these models, examining runs in the time sequence plot of residuals were used (Draper and Smith, 1966). From the residuals corresponding to the fit in Figure 5.6, P-value of the cumulative distribution of the three runs in samples of sizes (4, 6) was 0.048 (Draper and Smith, 1966). This suggested that the Gordon-Taylor equation was an adequate model for explaining T<sub>g</sub> behavior of the E<sub>2</sub> in ERS solid dispersions. For the residuals corresponding to the fit in Figure 5.7, P-value of the cumulative distribution of the three runs in samples of sizes (4, 7) was 0.033 (Draper and Smith, 1966). This suggested that the Kwei equation could also be used as a model for describing the T<sub>g</sub> behavior of these solid dispersions. In order to select the best model for predicting T<sub>g</sub> of this system, AIC values were calculated from the parameters obtained from each fit. The AIC values obtained from the fit of Gordon-Taylor and Kwei equations were 18.98 and 11.13, respectively. This indicated that Kwei equation fits the experimental data better than Gordon-Taylor equation.

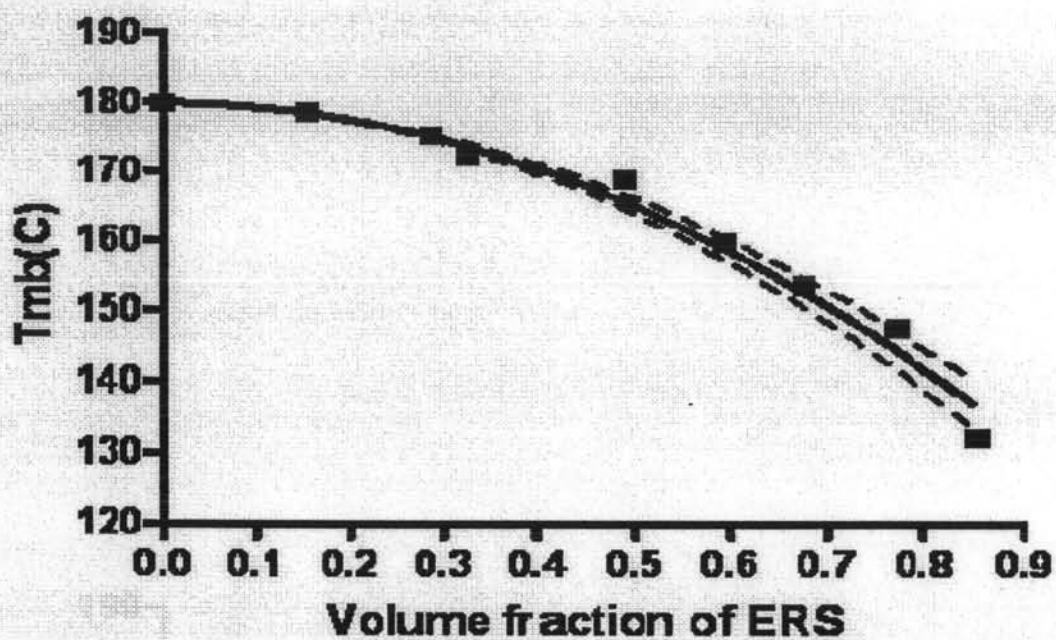


Figure 5.3 Fit of Nishi-Wang equation to experimental data: (■)  $T_{mb}$  of  $E_2$  in ERS solid dispersions obtained from experimental data; (---) Nishi-Wang equation

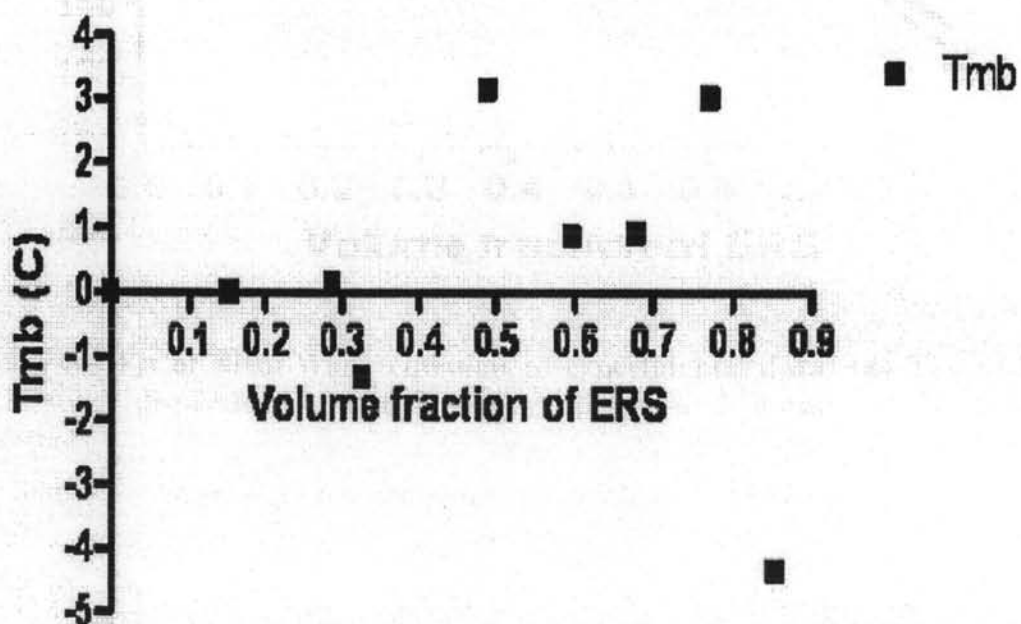


Figure 5.4 Residuals analysis corresponding to the fit of Nishi-Wang equation to experimental data

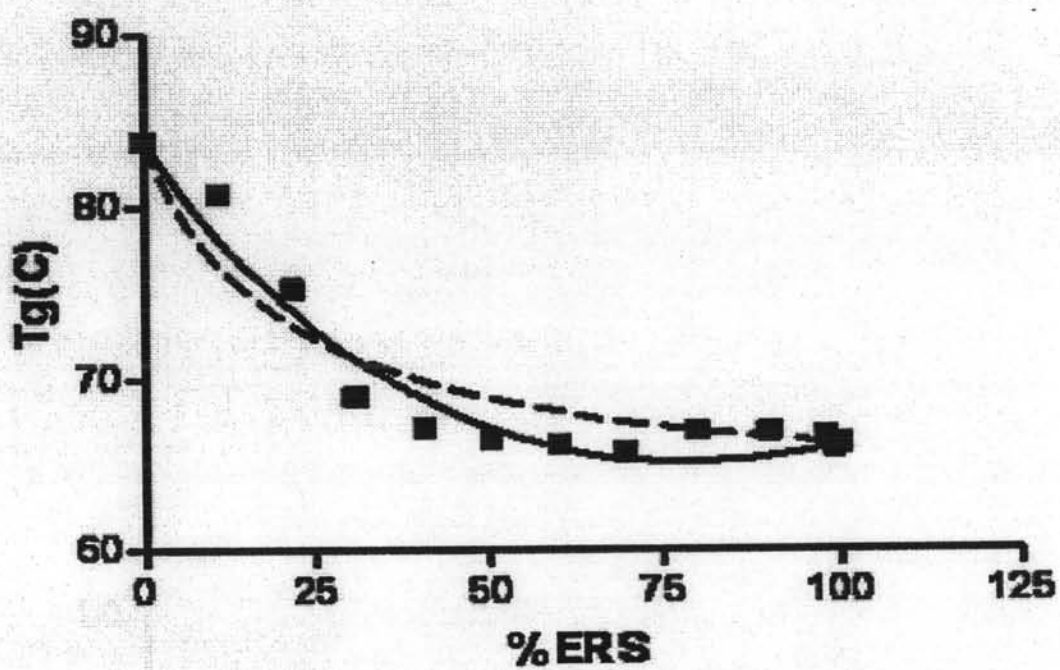


Figure 5.5  $T_g$  versus weight fraction of ERS curves based on experimental data (■); Gordon-Taylor equation (---); Kwei equation (—)

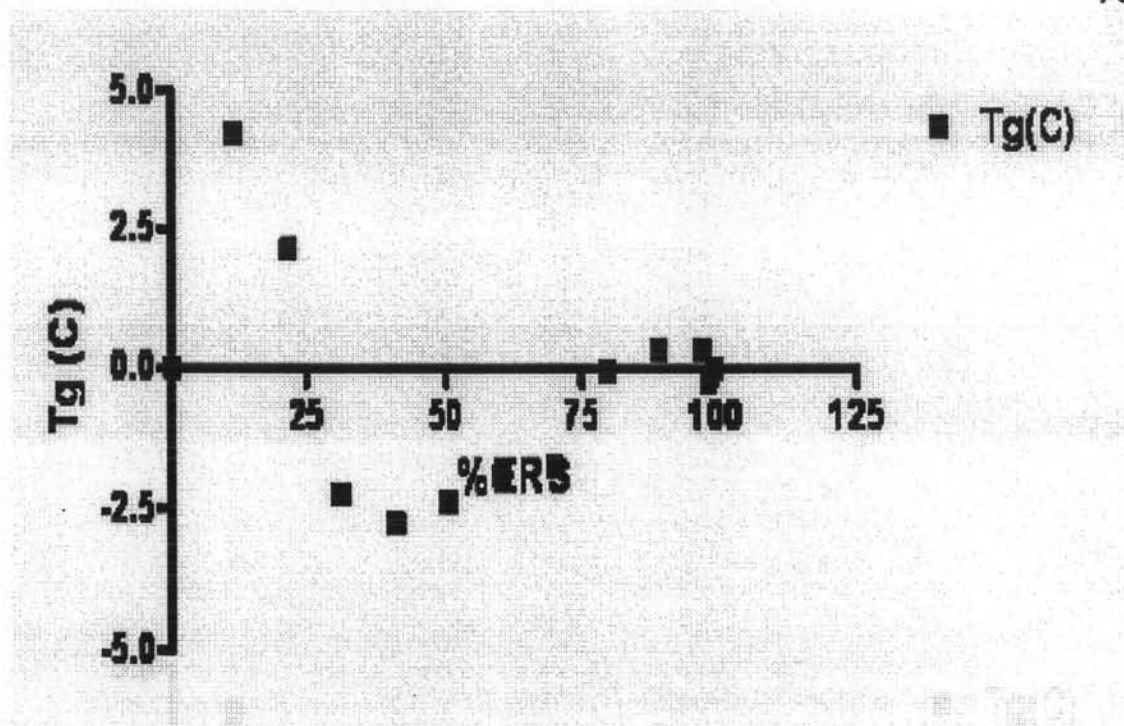


Figure 5.6 Residuals analysis corresponding to the fit of Gordon-Taylor equation to experimental data

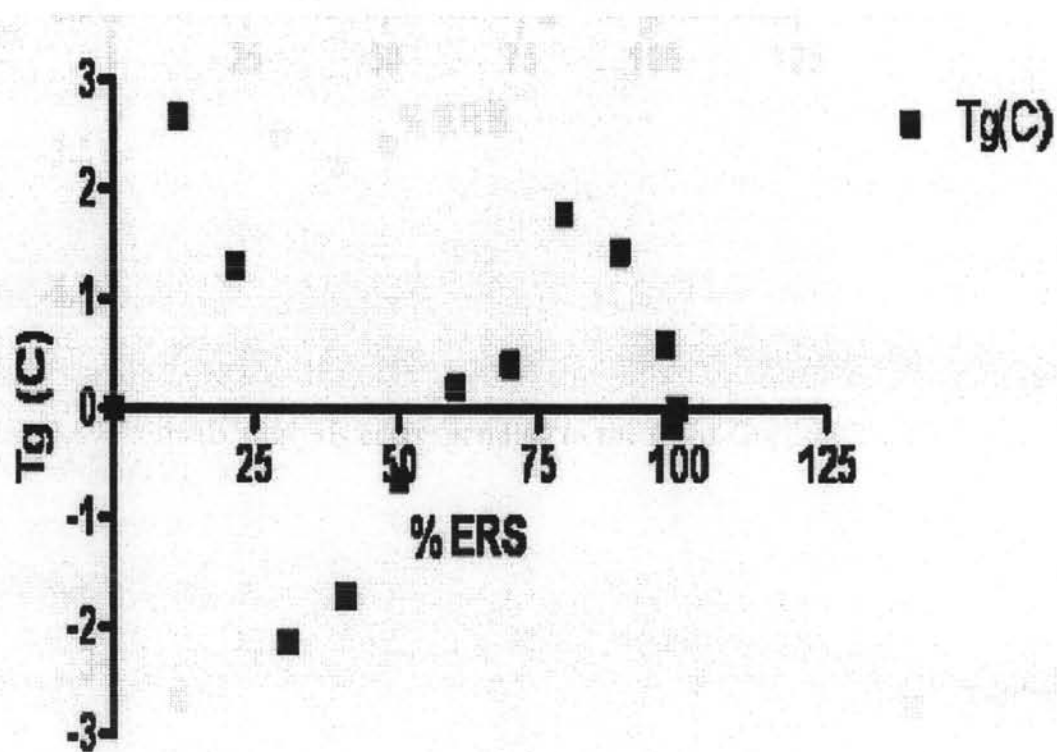


Figure 5.7 Residuals analysis corresponding to the fit of Kwei equation to experimental data

The estimate parameters,  $K$  and  $q$  values, determined from the Kwei fit were 0.4299 and  $-0.0017$ , respectively. The  $q$  value was a parameter corresponding to the strength of hydrogen bonding, reflecting the balance between breaking the self-associated hydrogen bonding and forming the inter-associated hydrogen bonding. The  $q$  value of the blend depended on an entropy change corresponding to the change in the number of hydrogen bonding. The negative  $q$  value of  $E_2$ -ERS system indicated that the fraction of the self-associated hydrogen bonding of  $E_2$  was broken to form weaker inter-associated hydrogen bonding with ERS. The estimate parameters,  $q$  value obtained from the Kwei fit and  $B$  value obtained from the Nishi-Wang fit, were not equal to zero indicating the interaction between  $E_2$  and ERS in the system.

### 5.3.7 FTIR Analysis of $E_2$ in ERS Solid Dispersions

FTIR spectra of  $E_2$ ,  $E_2$  in ERS solid dispersions, and ERS are displayed in Figure 5.8. Pure  $E_2$  showed very broad bands centered at  $3435.95$  and  $3232.06$   $\text{cm}^{-1}$  attributed to O-H stretching of hydroxyl group adjacent to C-17 and C-3 position in  $E_2$  chemical structure, respectively. These peaks were almost the same wavenumber as those reported by Barnett et al (1995). This O-H stretching range in the FTIR spectra was sensitive to the hydrogen-bonding formation corresponding to the broad band characteristic indicating the hydrogen-bonded hydroxyl group. When adding 25 percent by weight of ERS into solid dispersion, the broad hydrogen-bonded hydroxyl bands shifted to higher frequencies at  $3449.41$  and  $3246.95$   $\text{cm}^{-1}$  as shown in Figure 5.8 (A). However, the band at  $1732$   $\text{cm}^{-1}$  corresponding to the ester C=O stretching vibration of ERS as reported by Pignatello et al. (2002) did not change when adding  $E_2$  in solid dispersion. This suggested that hydroxyl group adjacent to C-17 and C-3 position of  $E_2$  could not engage in inter-associated hydrogen bonding with the ester C=O group in ERS solid dispersion analyzed at room temperature. In addition, an amine salt in the ERS structure was a quaternary ammonium salt which could have no N-H stretching vibrations (Silverstein, Bessler, and Morrill, 1991). Therefore, no evidence supported the inter-associated hydrogen bonding between  $E_2$  and ERS in solid dispersion analyzed by FTIR at room temperature. Due to water inclusion of  $E_2$  crystal, the broad bands in O-H stretching range were resulted from hydrogen-bonded hydroxyl group of  $E_2$  with water. The shifting of these bands to higher frequencies was a result of switching from the strong hydrogen-bonded hydroxyl group of  $E_2$  with water to the weak hydrogen-bonded hydroxyl group.

FTIR spectra of  $E_2$  and  $E_2$  in ERS solid dispersions heated from  $25$ - $175$   $^{\circ}\text{C}$  at  $5$   $^{\circ}\text{C}/\text{min}$ , and ERS are shown in Figure 5.9. The peak around  $3530$   $\text{cm}^{-1}$  corresponding to free hydroxyl group absorption (Kuo and Chang, 2001; Kuo et al., 2001) was observed in FTIR spectra of  $E_2$  and 75 % w/w  $E_2$  in ERS solid dispersion heated from  $25$ - $175$   $^{\circ}\text{C}$ . Because of water loss during heating process, the free hydroxyl group of  $E_2$  was observed. However, broad bands centered at  $3434.27$  and  $3245.29$   $\text{cm}^{-1}$  attributed to O-H stretching of hydroxyl group adjacent to C-17 and C-3 position in  $E_2$  chemical structure, respectively were still observed in FTIR spectrum of  $E_2$  heated from  $25$ - $175$   $^{\circ}\text{C}$ . The broad band centered at  $3434.27$   $\text{cm}^{-1}$  shifted to



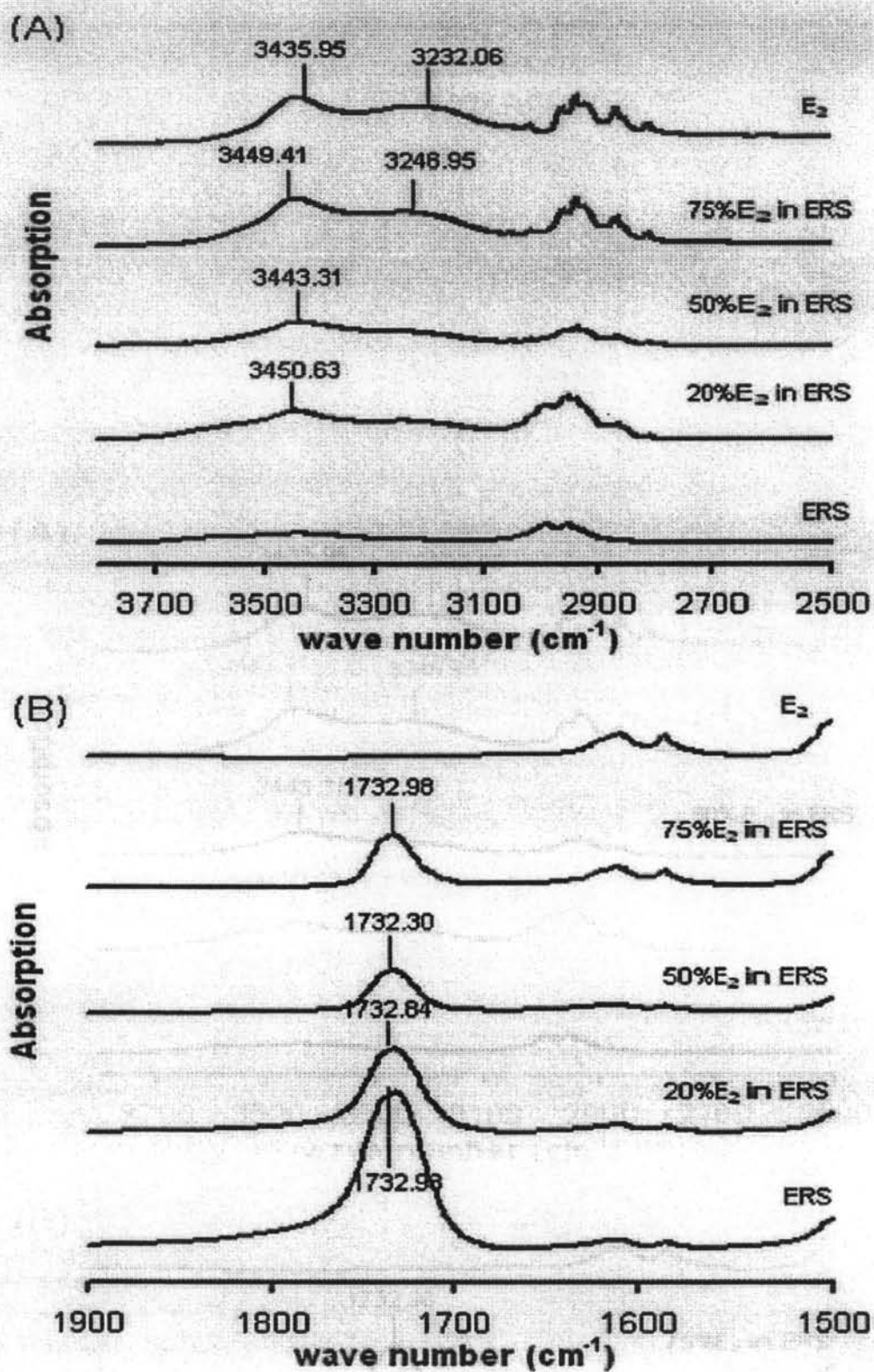


Figure 5.8 FTIR spectra of E<sub>2</sub> in ERS solid dispersions at concentration range of 0-100 % w/w recorded at room temperature in the range of 3800-2500 cm<sup>-1</sup> (A); 1900-1500 cm<sup>-1</sup> (B)

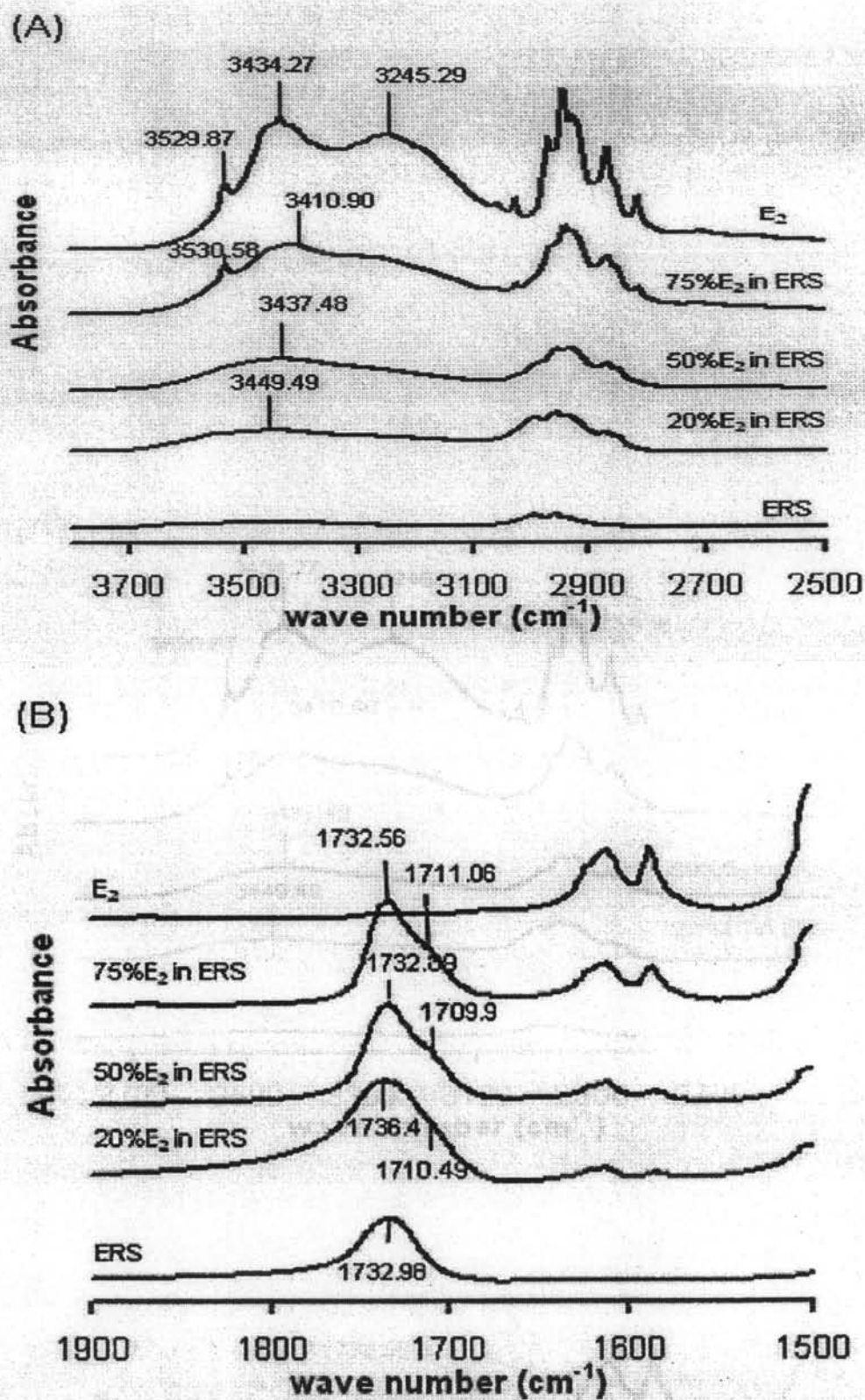


Figure 5.9 FTIR spectra of  $\text{E}_2$  in ERS solid dispersions at concentration range of 0-100 % w/w heated from 25-175  $^{\circ}\text{C}$  at 5  $^{\circ}\text{C}/\text{min}$  and recorded at room temperature in the range of 3800-2500  $\text{cm}^{-1}$  (A); 1900-1500  $\text{cm}^{-1}$  (B)

3410.90  $\text{cm}^{-1}$  corresponding to an occurrence of a shoulder of the ester C=O stretching band in FTIR spectrum of 75 % w/w  $E_2$  in ERS solid dispersion heated from 25-175  $^{\circ}\text{C}$ . The absorption around 1732  $\text{cm}^{-1}$  and the shoulder around 1710  $\text{cm}^{-1}$  corresponded to the free carbonyl group and the hydrogen-bonded carbonyl group, respectively (Kuo and Chang, 2001; Kuo et al., 2001). This suggested the switching from the intra-associated hydroxyl-hydroxyl bond to the inter-associated hydroxyl-carbonyl bond.

In FTIR spectra of 50 and 20 % w/w  $E_2$  in ERS solid dispersions heated from 25-175  $^{\circ}\text{C}$ , the peak around 3530  $\text{cm}^{-1}$  corresponding to free hydroxyl group disappeared and the broad bands centered at 3437.48 and 3449.49  $\text{cm}^{-1}$  were observed. This phenomenon corresponded to the shoulder around 1710  $\text{cm}^{-1}$  of the ester C=O stretching band of ERS. This suggested the inter-associated hydrogen bonding between free hydroxyl group of  $E_2$  and the ester C=O group of ERS when water inclusion of  $E_2$  crystal was removed. This result was in agreement with the negative  $q$  value obtained from Kwei equation confirming the interaction between  $E_2$  and ERS via hydrogen bonding. Moreover, this result agreed with the negative  $B$  value obtained from Nishi-Wang equation confirming the occurrence of interaction between  $E_2$  and ERS in molten state.

## 5.4 Conclusions

$E_2$  can exist in the amorphous form and at least three crystalline forms which are interconvertible by various thermal conditions. Each polymorphic form of crystalline  $E_2$  exhibited a XRPD pattern itself. The miscibility of  $E_2$  and ERS in solid dispersion could be determined by MTDSC. The criteria for indicating the miscibility of polymer blends, melting point depression and a single  $T_g$  point, could be applied for indicating the miscibility of  $E_2$  and ERS in solid dispersion. In molten state, the melting point depression as a function of composition of  $E_2$  in ERS solid dispersion was observed. This phenomenon could be explained by Nishi-Wang equation based on the Flory-Huggins theory. For a  $T_g$  behavior, the variation of  $T_g$  as a function of composition of  $E_2$  in ERS solid dispersion was observed when  $E_2$  existed in an amorphous form. The variation of  $T_g$  of  $E_2$  in ERS solid dispersion could be explained by Gordon-Taylor equation and its modified version equation, Kwei equation. Kwei equation was more appropriate than Gordon-Taylor equation to be used as a predictive model and the interaction between  $E_2$  and ERS via inter-associated hydrogen bonding could be observed in FTIR spectra.